

Three-Body Wear Behavior of AA1100/ZrC Metal Matrix Composites

¹M. Mastaniah and A. Chennakesava Reddy²

¹Research Scholar, Department of Mechanical Engineering, JNTU College of Engineering, Hyderabad, India

²Professor, Department of Mechanical Engineering, JNTU College of Engineering, Hyderabad, India

Abstract: In the present work, the AA1100- ZrC metal matrix composites were manufactured at 10% and 30% volume fractions of ZrC. The pin-on-disc wear test was conducted with different combinations of reinforcement, sliding distance, normal load, sliding speed. Based on the experimental results empirical models were established. The wear rate decreases with increase of vol.% ZrC in AA1100 alloy matrix. Three-body abrasive mechanism was predominant in the wear mechanisms of AA1100-ZrC metal matrix composites.

Keywords: AA1100, zirconium carbide, wear, sliding distance, normal load, sliding speed.

1. INTRODUCTION

Metal matrix composites have added advantages compared to the base material due to the presence of the reinforcements. Aluminum has a good thermal conductivity, less density and a high strength to weight ratio. Upon adding the material, the strength, stiffness, density and thermal/electrical conductivity of the composites improve considerably [1-19]. From the various studies on dry sliding wear of aluminum composites, effects of reinforcement size, volume fractions and morphology on the wear rate can be observed. Similarly, effects of various operation parameters on the dry sliding wear have been discussed in previous studies [20-27].

The objective of the present work was to evaluate the dry sliding wear behavior of AA1100/Zirconium carbide composites over a range of loads, sliding distances and sliding speeds.

2. MATERIALS METHODS

The matrix material was AA1100 alloy. The reinforcement material was zirconium carbide (ZrC) nanoparticles of average size 100nm. AA1100 alloy/ ZrC composites were fabricated by the stir casting process and low pressure casting technique with argon gas at 3.0 bar. The composite samples were given H18 heat treatment. The heat-treated samples were machined to get cylindrical specimens for the wear tests. The design of experiments was carried out as per Taguchi techniques [28, 29]. The levels chosen for the controllable process parameters are summarized in Table 1. The orthogonal array (OA), L9 was preferred to carry out wear tests experimentally (Table 2). A pin on disc type friction and wear monitor (ASTM G99) was employed to evaluate the friction and wear behavior of AA1100 alloy/ ZrC composites against hardened ground steel (En32) disc. To determine hardness before and after wear test, the Knoop hardness was conducted. Scanning electron microscopy analysis was also carried out to find consequence of wear test AA1100/ ZrC composite specimens.

Table 1: Control parameters and levels

Factor	Symbol	Level-1	Level-2	Level-3
Reinforcement, Vol.%	A	10	20	30
Load, N	B	20	30	40
Speed, m/s	C	2	3	4
Sliding distance, m	D	500	1000	1500

. Elastic modulus was computed as follows:

The upper-bound equation is given by

$$\frac{E_c}{E_m} = \left(\frac{1-v_v^{2/3}}{1-v_v^{2/3}+v_v} \right) + \frac{1+(\delta-1)v_p^{2/3}}{1+(\delta-1)(v_p^{2/3}-v_p)} \tag{1}$$

The lower-bound equation is given by

$$\frac{E_c}{E_m} = 1 + \frac{v_p-v_v}{\delta/(\delta-1)-(v_p+v_v)^{1/3}} \tag{2}$$

where, $\delta = E_p/E_m$, where, v_v and v_p are the volume fractions of voids/porosity and nanoparticles in the composite respectively and E_m and E_p is elastic moduli of the matrix and the particle respectively.

Table 2: Orthogonal array (L9) and control parameters

Treat No.	A	B	C	D
1	1	1	1	1
2	1	2	2	2
3	1	3	3	3
4	2	1	2	3
5	2	2	3	1
6	2	3	1	2
7	3	1	3	2
8	3	2	1	3
9	3	3	2	1

3. RESULTS AND DISCUSSION

The elastic stiffness and knoop hardness were increased with volume fraction of ZrC as shown in figure 1.

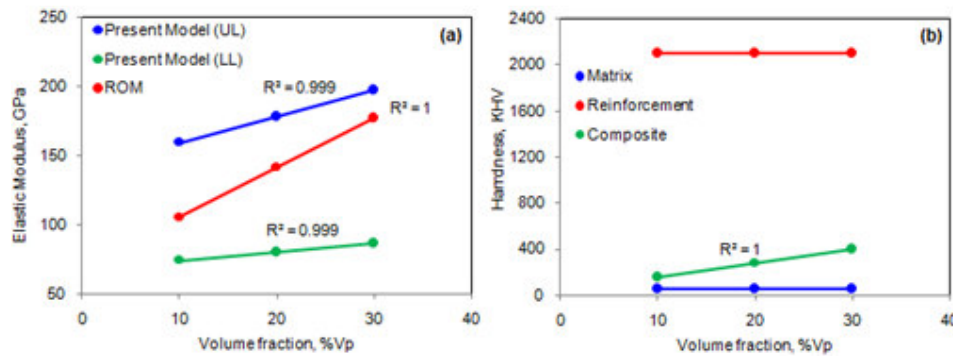


Figure 1: Mechanical Properties of AA1100/ZrC composites.

Table 3: ANOVA summary of the effective stress

Source	Sum 1	Sum 2	Sum 3	SS	v	V	F	P
A	21.24	20.30	19.50	0.51	1.00	0.51	8.90E+12	62.02
B	20.03	20.25	20.76	0.09	1.00	0.09	1.64E+12	11.47
C	20.58	20.51	19.95	0.08	1.00	0.08	1.40E+12	9.75
D	19.95	136.69	61.04	0.14	1.00	0.14	2.40E+12	16.76
e				0.00	4.00	0.00	1.00	0.00
T	81.80	197.75	121.25	0.82	8.00			100.00

Note: SS is the sum of square, v is the degrees of freedom, V is the variance, F is the Fisher's ratio, P is the percentage of contribution and T is the sum squares due to total variation.

3.1 Effect of volume fraction, Normal Load, Sliding Speed, Sliding distance on Wear Rate

All process parameters are acceptable as they satisfy Fisher's test at 90% confidence level. The analysis of variance (ANOVA) is presented in Table 3. The percent contribution indicates that the volume fraction of ZrC, contributes 62.02%. The normal load gives 11.47% of variation in the wear rate. The speed devotes 9.75% of variation in the wear rate. The sliding distance tends 16.76% of the total variation in the wear rate. The R-squared values of %reinforcement, normal load, sliding speed and

sliding distance are, respectively, 0.950, 0.883, 0.728 and 0.896. The trend of mean values obtained by Taguchi techniques is same as that of R-squared values. The wear rate was decreased with increase in volume fraction of ZrC in AA1100 alloy matrix (figure 2a). This is due to high hardness of ZrC as compared to soft matrix. The reduction of wear rate was due to increase of composite hardness as seen from figure 1c. An increase in wear rate is with increase of normal load applied on the test specimen (figure 2b). This effect was due to adhesion and increase in plastic deformation at the surface layers. The wear rate was decreased with increase of speed (figure 2c). At higher speeds an oxide layer was formed at the sliding interface. This, in turn, would avoid the direct metallic contacts resulting reduced wear rate. The wear rate increased with the sliding distance as shown in figure 2d.

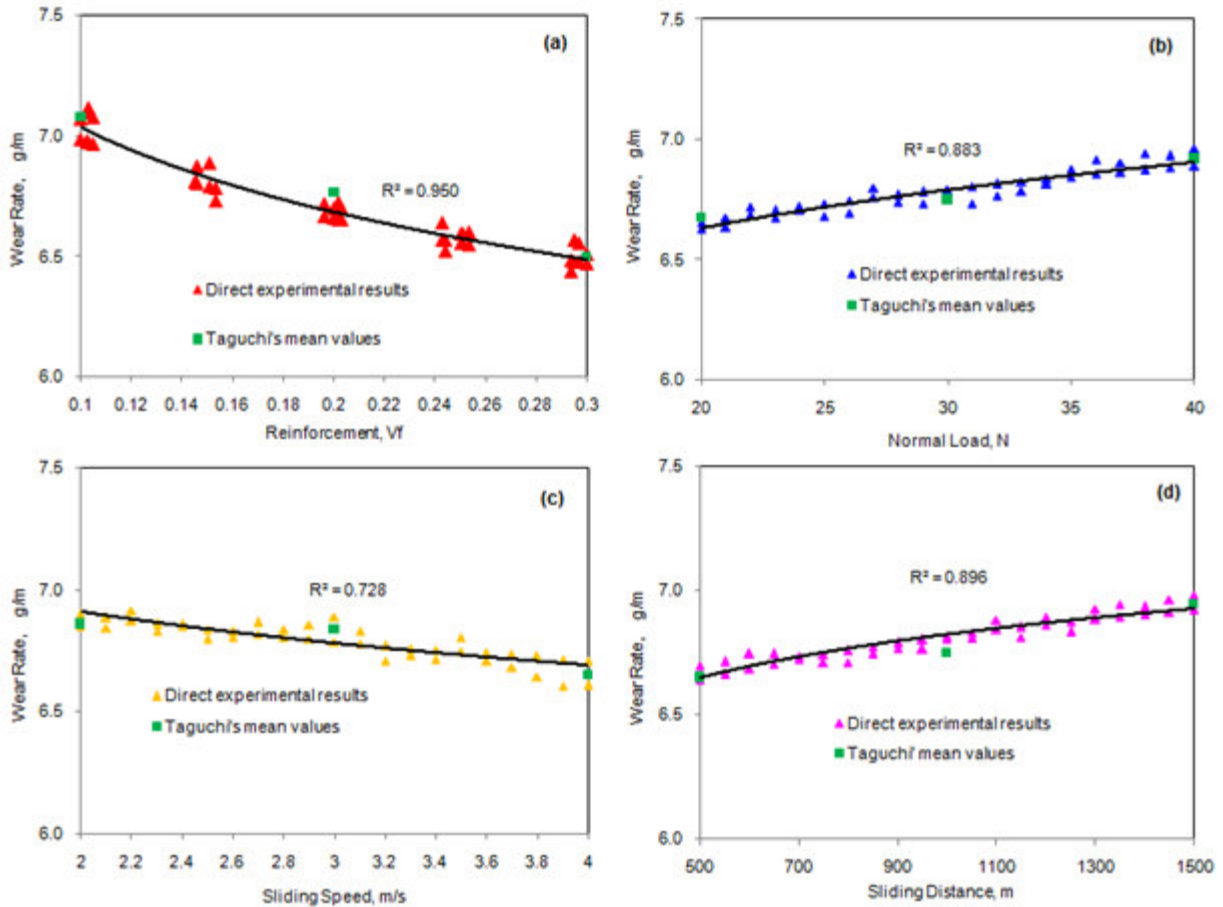


Figure 2: Influence of process parameters on wear rate.

The mathematical relation between wear and contact time is given by

$$W_{rp} = 5.933v_f^{-0.05} \quad (3)$$

$$W_{rf} = 5.57F^{0.058} \quad (4)$$

$$W_{rn} = 7.131N^{-0.04} \quad (5)$$

$$W_{rd} = 5.263d^{0.037} \quad (6)$$

where,

W_{rp} is the wear rate due to vol.% of reinforcement (v_f), g/m

W_{rf} is the wear rate due to normal load (F), g/m

W_{rn} is the wear rate due to speed (N), g/m

W_{rd} is the wear rate sliding distance (d), g/m.

3.2 Consequence of Wear in AA1100/ZrC Composites

It is important to identify the consequence of wear in AA1100/ZrC composites. The reason of post-wear evaluation is to focus the changes that are brought in the worn specimens in terms of mechanical properties, microstructure, and worn-surface pattern. The change in hardness of the worn specimens is shown in figure 3. It can be seen that the hardness values increase after wear test. The increase in hardness in the worn specimens may be attributed to the work hardening and the frictional temperature. When the load was increased, the plastic deformation would take place in the AA1100 alloy matrix which could adhere to the steel disc and consequently, resulting the conditions of adhesive wear as shown in figure 4. If the wear is caused by a hard particle (ZrC) trapped between the rubbing surfaces, the phenomenon is called three-body abrasive wear. The particle removal was higher in the composites having 30% Si₃N₄ than in the composites having 10% ZrC.

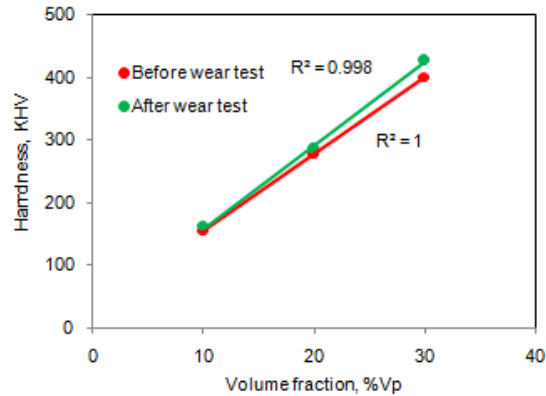


Figure 3: Hardness of AA1100/ZrC composites after wear test.

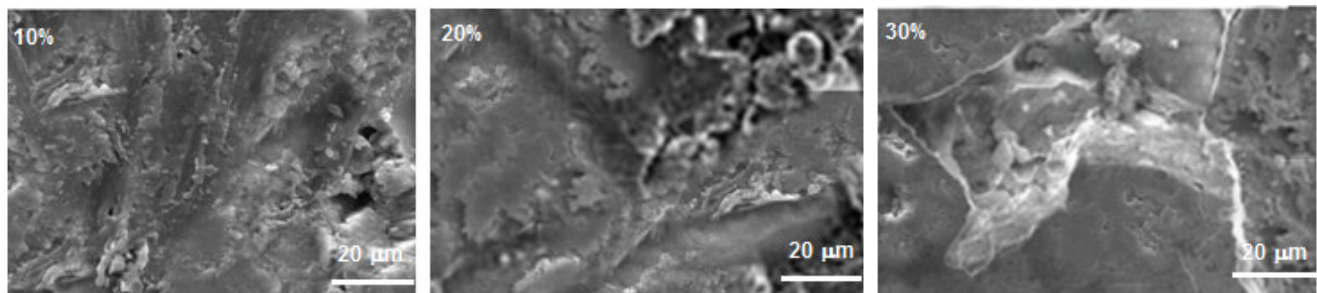


Figure 4: Adhesive and abrasive wear of AA1100/ZrC.

4. CONCLUSION

The study on the wear behavior of AA1100/ZrC composites as the function of vol.% of ZrC, normal load, sliding speed and sliding distance using Taguchi's design of experiments was carried out successfully. The wear resistance increases with increase of vol.% ZrC nanoparticles in AA1100 alloy matrix. The wear rate increases with increase in normal load and sliding distance. The dominant mechanisms of adhesive wear and three-body abrasive wear.

REFERENCES

1. A. C. Reddy, Mechanical properties and fracture behavior of 6061/SiC_p Metal Matrix Composites Fabricated by Low Pressure Die Casting Process, *Journal of Manufacturing Technology Research*, 1, 2009, pp.273-286.
2. A. C. Reddy, Tensile properties and fracture behavior of 6063/SiC_p metal matrix composites fabricated by investment casting process, *International Journal of Mechanical Engineering and Materials Sciences*, 3, 2010, pp.73-78.
3. A. C. Reddy and B. Kotiveerachari, Effect of aging condition on structure and the properties of Al-alloy / SiC composite, *International Journal of Engineering and Technology*, 2, 2010, pp.462-465.
4. A. C. Reddy and B. Kotiveerachari, Influence of microstructural changes caused by ageing on wear behaviour of Al6061/SiC composites, *Journal of Metallurgy & Materials Science*, 53, 2011, pp. 31-39.
5. A. C. Reddy, Tensile fracture behavior of 7072/SiC_p metal matrix composites fabricated by gravity die casting process, *Materials Technology: Advanced Performance Materials*, 26, 2011, pp. 257-262.

6. A. C. Reddy, Influence of strain rate and temperature on superplastic behavior of sinter forged Al6061/SiC metal matrix composites, *International Journal of Engineering Research & Technology*, 4, 2011, pp.189-198.
7. A. C. Reddy, Evaluation of mechanical behavior of Al-alloy/SiC metal matrix composites with respect to their constituents using Taguchi techniques, *i-manager's Journal of Mechanical Engineering*, 1, 2011, pp.31-41.
8. A. C. Reddy and Essa Zitoun, Matrix al-alloys for alumina particle reinforced metal matrix composites, *Indian Foundry Journal*, 55, 2009, pp.12-16.
9. A. C. Reddy and Essa Zitoun, Tensile behavior of 6063/Al₂O₃ particulate metal matrix composites fabricated by investment casting process, *International Journal of Applied Engineering Research*, 1, 2010, pp.542-552.
10. A. C. Reddy and Essa Zitoun, Tensile properties and fracture behavior of 6061/Al₂O₃ metal matrix composites fabricated by low pressure die casting process, *International Journal of Materials Sciences*, 6, 2011, pp.147-157.
11. A. C. Reddy, Strengthening mechanisms and fracture behavior of 7072Al/Al₂O₃ metal matrix composites, *International Journal of Engineering Science and Technology*, 3, 2011, pp.6090-6100.
12. A. C. Reddy, Evaluation of mechanical behavior of Al-alloy/Al₂O₃ metal matrix composites with respect to their constituents using Taguchi, *International Journal of Emerging Technologies and Applications in Engineering Technology and Sciences*, 4, 2011, pp. 26-30.
13. A. C. Reddy, Sliding Wear and Micromechanical Behavior of AA1100/Titanium Oxide Metal Matrix Composites Cast by Bottom-Up Pouring, 7th International Conference on Composite Materials and Characterization, Bangalore, 11-12 December 2009, 205-210.
14. A. C. Reddy, Fracture behavior of brittle matrix and alumina trihydrate particulate composites, *Indian Journal of Engineering & Materials Sciences*, 9, 2002, pp.365-368.
15. A. C. Reddy, Wear and Mechanical Behavior of Bottom-Up Poured AA4015/Graphite Particle-Reinforced Metal Matrix Composites, 6th National Conference on Materials and Manufacturing Processes, Hyderabad, 8-9 August 2008, pp. 120-126.
16. A. C. Reddy, S. Sundararajan, Influences of ageing, inclusions and voids on the ductile fracture mechanism of commercial Al-alloys, *Journal of Bulletin of Material Sciences*, 28, 2005, pp. 75-79.
17. A. C. Reddy, S. Madahava Reddy, Evaluation of dry sliding wear characteristics and consequences of cast Al-Si-Mg-Fe alloys, *ICFAI Journal of Mechanical Engineering*, 3, 2010, pp.1-13.
18. A. C. Reddy, M. Vidya Sagar, Two-dimensional theoretical modeling of anisotropic wear in carbon/epoxy FRP composites: comparison with experimental data, *International Journal of Theoretical and Applied Mechanics*, 6, 2010, pp. 47-57.
19. A. S. Goud, A. C. Reddy, Evaluation of Nanoparticle Fracture in MgO Reinforced Aluminum matrix composites, 3rd International Conference on Modern Materials and Manufacturing, New Delhi, 9-10 December 2011, pp. 320-324.
20. A. S. Goud, A. C. Reddy, Interface Failure Analysis of TiB₂ Reinforced Aluminum Alloy Matrix Composites, 3rd International Conference on Modern Materials and Manufacturing, New Delhi, 9-10 December 2011, pp. 325-328.
21. M. S. Ramgir, A. C. Reddy, Control of B₄C Reinforced Particulates on Dry Wear Resistance of AA2024/B₄C Composites, 3rd International Conference on Modern Materials and Manufacturing, New Delhi, 9-10 December 2011, pp. 336-340.
22. V. K. Reddy, A. C. Reddy, Mathematical Models for Dry Wear of H18 Heat Treated AA1100/TiB₂ Composites, 3rd International Conference on Modern Materials and Manufacturing, New Delhi, 9-10 December 2011, pp. 341-346.
23. M. Mastanaiah, A. C. Reddy, Abrasive Wear of AA3003/ZrC Composites, 3rd International Conference on Modern Materials and Manufacturing, New Delhi, 9-10 December 2011, pp. 347-351.
24. R. G. Math, A. C. Reddy, Unlubricated Sliding of AA4015/TiB₂ Metal Matrix Composites, 3rd International Conference on Modern Materials and Manufacturing, New Delhi, 9-10 December 2011, pp. 352-356.
25. M. L. T. Guo, C. Y. A. Tsao, Tribological behavior of self-lubricating aluminum/SiC/graphite hybrid composites synthesized by the semi-solid powder-densification method, *J. Composites Sci. Tech.*, 60, 2000, pp.65-74.
26. T. R. Chapman, D. E. Niez, R. T. Fox, R. Fawcett, Wear-resistant aluminum-boron-carbide cermets for automotive brake applications, *Journal of Wear*, 236, 1999, pp. 81-87.
27. A. Chennakesava Reddy, V.M. Shamraj, Reduction of cracks in the cylinder liners choosing right process variables by Taguchi method, *Foundry Magazine*, 10, 1998, pp. 47-50.
28. A. Chennakesava Reddy, V.S.R. Murti, S. Sundararajan, Control factor design of investment shell mould from coal flyash by Taguchi method, *Indian Foundry Journal*, 45, 1999, pp. 93-98.