Correlation of Surface Profiles and Worn Surfaces of AA6061/Graphite Metal Matrix Composites

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Abstract: In the present work, the AA6061alloy/graphite metal matrix composites were manufactured at 10% and 30% volume fractions of graphite. The pin-on-disc wear test was carried out with different combinations of reinforcement, sliding distance, normal load, sliding speed. Based on the experimental results surface profiles of worn surfaces were constructed and correlated.

Keywords: Metal matrix composite, surface profiles, AA6061 alloy, graphite, wear, sliding distance, normal load, speed.

1. INTRODUCTION

Metal matrix composites reinforced with nano-particles are being investigated worldwide in recent years, owing to their promising properties suitable for a large number of functional and structural applications. In the literature, different kinds of matrix metals have been coupled with several types of nanoscale phases [1-17]. Aluminum-graphite composites are made by reinforcing graphite nanoparticle in molten aluminum alloy. At present the composite is being extensively field tested across world as a replacement for expensive conventional bearing alloys such as phosphor bronze for pistons, piston rings and cylinder liners [18]. Improvements in tribological properties are accompanied by a decrease in the strength and ductility of aluminum-graphite composites [19]. The dry wear characteristics of aluminum- graphite composites are influenced by the transport of graphite from the bulk to the mating surface during sliding and the plastic deformation of the subsurface. Graphite addition to aluminum alloy deteriorates the wear properties. High wear can occur in mechanisms which operate in conditions of dry friction [20-23].

The purpose of this research work was to correlate surface profiles with worn surfaces AA6061/graphite metal matrix composites. The wear tests were conducted on pin-on-disc equipment. The design of experiments was based on Taguchi techniques [24, 25].

2. MATERIALS AND METHODS

AA6061 alloy/graphite composites were fabricated by the stir casting process. The T8 heat-treated samples were machined to get cylindrical specimens of 10 mm diameter and 30 mm length for the dry wear tests. The levels chosen for the controllable wear parameters are précised in Table 1. The orthogonal array, L9 was ideal to carry out wear experiments (Table 2). A pin-on-disc wear monitor (ASTM G99) was employed to assess the wear behavior of AA6061 alloy/graphite composite specimens against hardened ground steel (En32) disc (figure 1a). Knoop microhardness was conducted before and after wear tests. The surface roughness of worn samples was measured using Talysurf surface roughness tester as shown in figure 1b.



Figure 1: Tests carried out in the present work: (a) Pin-on-disc wear test and (b) Surface roughness test.

Factor	Symbol	Level-1	Level-2	Level-3
Reinforcement, Vol.%	Α	10	20	30
Load, N	В	10	20	30
Speed, m/s	С	2	3	4
Sliding distance, m	D	200	400	600

Table 1: Wear parameters and levels

Table 2: Orthogonal array (L9) and control parameters

Treat No.	Α	В	С	D
1	1	1	1	1
2	1	2	2	2
3	1	3	3	3
4	2	1	2	3
5	2	2	3	1
6	2	3	1	2
7	3	1	3	2
8	3	2	1	3
9	3	3	2	1

3. RESULTS AND DISCUSSION

All process parameters are acceptable as they satisfy Fisher's test at 90% confidence level. The analysis of variance (ANOVA) is presented in Table 3. The graphite volume fraction (A) accords 53.52% of variation the wear rate. The normal load (B) affords 12.11% of variation in the wear rate. The sliding speed (C) allows 0.56% of variation in the wear rate. The sliding distance (D) cedes 33.80% of variation in the wear rate. The R-squared values of %reinforcement, normal load, sliding speed and sliding distance are, respectively, 0.999, 0.971, 0.279 and 0.991. The trend of mean values obtained by Taguchi techniques is same as that of R-squared values. The influence of sliding speed is nearly negligible on the wear rate of AA6061 alloy/graphite composites.

Table 3: ANOVA summary of the effective stress

Source	Sum 1	Sum 2	Sum 3	SS	v	V	F	Р
А	1.35	1.19	1.05	1.50E-02	1	1.50E-02	9.02E+13	34.70
В	1.13	1.16	1.30	5.49E-03	1	5.49E-03	3.30E+13	12.68
С	1.15	1.22	1.22	1.09E-03	1	1.09E-03	6.54E+12	2.52
D	1.01	0.49	3.59	2.17E-02	1	2.17E-02	1.30E+14	50.10
e				-6.66E-16	4	-1.67E-16	1.00	0.00
Т	4.64	4.06	7.16	4.33E-02	8			100.00

Note: SS is the sum of square, v is the degrees of freedom, V is the variance, F is the Fisher's ratio, P is the percentage of contribution and T is the sum squares due to total variation.

The wear rate was decreased with increase in volume fraction of graphite in AA6061 alloy matrix as shown in figure 2a. For a soft-hard material sliding interaction the input energy is dissipated in giving rise to intense plastic deformation in the soft substrate under dry wear conditions. The strain hardening was high in AA6061 alloy matrix and low graphite (10%) composite, while the high graphite (30%) composite shows comparatively less hardening. Under these circumstances the less ductile material tends to produce wear fragments which become more readily detached from the parent body because of fracture and at a faster rate than those of the more ductile material. Thickness of the wear fragment decreases with increasing content of graphite in the composite. As seen from figures 2b, an increase in wear rate is with increase of normal load applied. Due to increased applied load, the deformed AA6061 alloy matrix disengages from the pin's surface. The deformed matrix material adheres to the steel disc surface, consequently raising the wear rate. Under all loads the wear track on the disc became smeared with a layer of AA6061 alloy matrix within a short period. Subsequently, sliding took place between this transferred layer and the test specimen, thus promoting seizure. A number of seizure particles, the size of which decreased with the graphite content of the composite, adhered to the track. There is no significant variation in wear with sliding speed for the AA6061/graphite composites (figure 2c). The wear rate increases with sliding distance as shown in figure 2d. For long sliding distances, the shear stress at the interface exceeds the shear strength of the particle/matrix interface resulting particle debonding. Disintegration of the surface layers thus proceeds at faster rates with increasing sliding distance.



Figure 2: Wear rate as a function of (a) % reinforcement, (b) applied load, (c) sliding speed and (d) sliding distance.

The surface contour of AA6061/10% graphite composites is due to removal of the matrix alloy from the pin's surface and movement of matrix alloy onto the groove edges. Thickness of the wear fragment is high with 10% graphite in the composite. In the composites having 10% graphite content, the stain hardening and plastic deformation are also high which results the surface profile as shown in figure 4. The surface roughness contours of AA6061/20% graphite and AA6061/30% graphite composites represent the improvement of wear resistance due to addition of graphite nanoparticles (figure 4). But, in the composites having 30% graphite, the particle debonding is high causing the rough surface profile. The surface profiles coincide with the micrographs of AA6061/graphite composites.

4. CONCLUSIONS

The surface profiles of AA6061/graphite composites were successfully correlated with micrographs of the worn surfaces. The wear resistance increases with increase in volume fraction of graphite nanoparticles in AA6061 alloy matrix due to increased hardness of the composites. The sliding speed has no significance on the wear rate of AA6061/graphite composites.

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Figure 4: Surface profiles and worn surfaces of AA6061/graphite composites after wear test.

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